# Quantum-size effects in sub 10-nm fin width InGaAs FinFETs

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# Abstracts

InGaAs FinFETs with sub-10 nm fin widths were fabricated for the first time using precision dry etching and digital etch. We find that the threshold voltage,  $V_T$ , becomes highly sensitive to the fin width,  $W_f$ , in the sub-10 nm  $W_f$  range. 2D Poisson-Schrodinger simulations suggest that this is due to quantization effects. We also show that in the quantum regime, a sidewall slope below 85<sup>o</sup> significantly reduce  $V_T$  variation at the same drawn dimensions.

#### Introduction

InGaAs has emerged as the most promising n-channel material for sub-10 nm CMOS [1]. In this dimensional range, only high aspect-ratio 3D transistors with a fin or nanowire configuration can deliver the necessary performance [2]. However, due to the low effective mass of InAs-rich III-V compounds, quantum effects are predicted to introduce great sensitivity of device characteristics to the fin width as it scales down to 7 nm and below [3]. In this work we fabricate InGaAs fin MOSFETs featuring fin widths in the sub-10 nm range and explore the role of quantum effects on  $V_T$ . We do indeed experimentally confirm the enhanced sensitivity of  $V_T$  to  $W_f$ , as compared with Si FinFETs with similar fin dimensions. The results have significant implications for future CMOS manufacturing.

### **Process technology**

The FinFET process flow is illustrated in Fig. 1. This process closely follows that of [4]. In essence, this is an n-type doped-channel FinFET in which the gate modulates the electron concentration from the fin sidewalls only. The starting material consists of a 50 nm thick  $In_{0.53}Ga_{0.47}As$  active layer Si-doped at  $10^{18}$  cm<sup>-3</sup> on an InAlAs buffer layer grown by MBE on SI-InP.

A novel reactive-ion etching process that utilizes a BCl<sub>3</sub>/SiCl<sub>4</sub>/Ar chemistry has been used to create the fins [5]. HSQ defined by e-beam lithography is used as mask. Fins as narrow as 15 nm with an aspect ratio of 10 have been obtained [5]. 5 cycles of digital etch were used to reduce the fin width and smooth the sidewalls [6]. Our digital etch process consists of a self-limited plasma-oxidation/wet oxide etch sequence. In

sample SW80 (SW90), the wet etch was done using  $H_2SO_4$  (diluted-HCl). In both cases, digital etch reduces the fin width by ~2 nm/cycle while preserving its shape [5]. In our process, the sidewall slope at the top of the fin is related to the fin height and can be controlled by the fin etching time. Two different fin sidewall slopes were fabricated (Fig. 2). Sample SW90 has a fin height of 130 nm and vertical sidewalls at the top 50 nm. Sample SW80 has a fin height of 60 nm with a sidewall slope of ~80-85°. In this sample,  $W_f$  represents the dimension at the top of the fin. The bottom of the channel is enlarged by ~8 nm.



Fig. 1: (a) InGaAs FinFET process flow. (b) Finished device



Fig. 2: Left: SW90 fins with vertical sidewalls. Right: SW80 fin with 80-85<sup>o</sup> sidewalls.

Immediately after digital etch, the sample is loaded into an ALD reactor for gate dielectric deposition. We used 5.5 nm of composite  $Al_2O_3/HfO_2$  with EOT ~3 nm. Gating from the top facet of the fins is suppressed by leaving in place the HSQ mask (>10 nm thick) used to pattern the fins. The gate metal consists of sputtered Mo patterned by RIE. The gate covers the entire length of the fin and overlaps with the extrinsic regions. Evaporated Mo/Ti/Au is used for source and drain contacts

and pads. The final step is annealing in forming gas for 30 min at 400°C.

FIB cross-section of finished devices with sub-10 nm fins are shown in Fig. 3a-b. A typical device consists of 100 fins, 3  $\mu$ m long (Fig. 4). W<sub>f</sub> ranges between 5 and 35 nm.

# Results

Output and subthreshold characteristics of a  $W_f=7$  nm SW90 device are shown in Figs. 5-6. The current is normalized by the number of fins (100). Well-behaved long-channel behavior is obtained with negligible DIBL. This indicates excellent electrostatic charge control on the fin sidewalls.

Room temperature (RT) subthreshold characteristics at  $V_{DS}$ =50 mV for devices with different  $W_f$  are shown in Fig. 7a-b. A progressive positive  $V_T$  shift as  $W_f$  scales down is observed in both devices.  $V_T$  extracted at  $I_D$ =5 nA/fin is graphed in Fig. 7c-d. Sample SW90 exhibits an enhanced  $V_T$  dependence on  $W_f$  for values of  $W_f$ < 10 nm that is not observed in the SW80 sample.



Fig. 3 FIB cross-sections of finished devices with **sub**-10 nm fins: (a) SW90 with vertical sidewalls, (b) SW80 with sloped sidewalls.

In order to understand the origin of this enhanced  $W_f$  dependence, we have carried out simultaneous I-V and C-V measurements at 90K at high frequency (500 KHz). This allows us to extract the mobility and sheet electron concentration while suppressing the impact of interface states. The temperature dependence of subthreshold characteristics of a typical device is shown in Fig. 8.  $V_T$  at 90K for both sets of devices is shown in Fig. 7c-d. Besides a nearly constant shift in  $V_T$ , low temperature brings a reduction in the  $V_T$  dependence for wide  $W_f$ . We attribute this to the reduced impact of interface states. However, the enhanced  $V_T$  dependence on  $W_f$  for very narrow SW90 devices remains at 90K.



Fig. 4: SEM micrograph of a finished device.

The combination of C-V and I-V measurements (Figs. 9a and c) allows the extraction of the mobility, as shown in Fig. 9b at T=90K. For high enough carrier concentrations, a rather flat dependence of  $\mu$  on n is observed as expected in a doped channel.



Fig. 5: Output characteristics of  $W_i$ =7 nm SW90 device.  $V_{cs}$  runs from 0 to 0.6 V in 0.1 V steps.

We estimate the electron concentration (n) in the subthreshold regime by using the mobility data obtained from CV-IV and the subthreshold current characteristics at 90 K. From this (Fig. 9d), we have extracted the threshold voltage,  $V_{Tn}$ , defined at a fixed carrier concentration of  $5 \cdot 10^5$  cm<sup>-1</sup> (Fig. 10). The SW80 data is characterized by a nearly constant slope while the SW90 samples with W<sub>f</sub><10 nm continue to exhibit an enhanced W<sub>f</sub> dependence.



Fig. 6: Subthreshold characteristics of W<sub>f</sub>=7 nm SW90 device.



Fig 7: Room temperature subthreshold characteristics of devices with different  $W_f$  of SW90 (a) and SW80 (b).  $V_T$  as a function of  $W_f$  for SW90 (c) and SW80 (d) at room temperature and 90 K.

#### 2D Poisson-Schrodinger simulations and discussion

In order to understand these experimental results, we have performed self-consistent 2D Poisson-Schrodinger simulations using Nextnano. Fig. 11 graphs  $V_{Tn}$  as extracted from these simulations at a fixed carrier concentration of  $5x10^5 \text{ cm}^{-1}$ . On both types of devices, there is a clear steeper dependence of  $V_{Tn}$  on  $W_f$  in the sub-10 nm range. The dependence is more prominent in the structure with vertical sidewalls.

The origin of this can be understood by mapping the spatial distribution of electrons across the fins (Fig. 12). In the SW80 sample, the electron distribution peaks at the wide bottom of the fin while in the SW90 devices it has a narrower lamella shape that spans a large portion of the fin height. The enhanced steepness of  $V_{Tn}$  for  $W_f < 10$  nm in both samples

arises from quantum effects. This can be seen by the classic  $V_T$  calculation shown in Fig. 11 that is successful in capturing the evolution of  $V_T$  in wider devices. The classic  $V_T$  is defined as the gate voltage that, within the depletion approximation, pinches off the fin. The P-S simulated  $V_{Tn}$  (shifted by 200 mV due to work function difference not accounted for in the simulations) is compared with the experimental results in Fig. 10 (full lines). The agreement gives us confidence in our interpretation.



Fig 8: Transfer characteristics of  $W_r$ =7 nm SW90 device at different temperatures.  $V_{DS}$ =50 mV.

When all the data are plotted vs. the average  $W_f$ , the quantum size effects in both samples are equivalent (inset of Fig. 10).

The relevance of this work is highlighted by comparing the  $V_T$  dependence expected from P-S simulations in fins made out of different materials (Fig. 13). As a result of their low effective mass, quantum confinement effects in GaAs, InGaAs and InAs are much more significant than in Si. At  $W_r=5$  nm, the sensitivity of  $V_T$  to  $W_r$  is 4 times higher for InGaAs than for Si. This implies that tight fin width control will be required in future InGaAs FinFETs and Trigate MOSFETs. In addition, the use of slanted sidewalls might help alleviate this issue though the consequences for charge control and short-channel effects need to be understood.

### Conclusions

We demonstrate for the first time InGaAs FinFETs with sub-10 nm fin widths. In this dimensional range, we observe a strong dependence of  $V_{\tau}$  on fin width that is due to quantum confinement effects. This is far enhanced above Si-based devices due to the low effective mass of electrons in InGaAs. This has important implications for the design and manufacturing of future InGaAs MOSFETs.



Fig 9: C-V (a) and I-V (c) characteristics of SW90 devices at  $V_{DS}$ =50 mV and T=90K. These are used to extract mobility (b). These measurements yield n-V<sub>GS</sub> curves (d).



Fig 10:  $V_{Tn}$  (defined at n=5.10<sup>5</sup> cm<sup>-1</sup>) vs.  $W_f$ . Lines: theory from Fig. 11 shifted by 0.2 V. Inset shows the same data plotted vs. average fin width.



Fig 11:  $V_{Tn}$  as a function of  $W_f$  for SW90 and SW80 fins as extracted from 2D P-S simulations. Also included is  $V_T$  calculated using classic model.



Fig 12: Electron density distribution at  $V_{\text{T}}$  for SW90 (left) and SW80 (right)





Fig 13: Simulated  $V_{Tn}$  vs.  $W_f$  for FinFETs with vertical sidewalls as obtained from 2D P-S simulations for different III-V semiconductors and comparison with Si.

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